# $T_N$ Dependence on Hole Concentration in the La<sub>2-x</sub> Sr<sub>x</sub> CuO<sub>4+ $\delta$ </sub> System

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Hole concentration, h, was determined by using the coulometric titration technique for the powdered samples of  $\text{La}_{2-x}\text{Sr}_x\text{CuO}_{4+\delta}$  ( $0 \le x \le 0.02$ ) annealed in flowing 1 atm  $N_2$  and 1 atm  $O_2$ . From the temperature dependence of magnetic susceptibility we found that the Néel temperature,  $T_N$ , decreases steeply with increasing h and that  $T_N(h)$  can be fitted in the following expression  $T_N(h)/T_N(0)=1-\{h/h_c\}^n$  where  $T_N(0)$ ,  $h_c$  and n are constants. This power-law dependence might be explained by finite size scaling theory.

2D antiferromagnet, Néel temperature, hole concentration, finite size scaling, high- $T_{\rm c}$  superconductor

### §1. Introduction

One of the most interesting features of cuprate high- $T_c$  superconductors is that the parent compounds are two-dimensional antiferromagnetic insulators.<sup>1)</sup> The antiferromagnetic ordering vanishes steeply with doping holes and further doping causes superconductivity.<sup>2)</sup> In order to reveal a correlation between antiferromagnetic ordering and doped holes,  $\text{La}_{2-x}\text{Sr}_x\text{CuO}_{4+\delta}$  is one of the suitable systems, because the structure is most simple, i.e. the structure has only one  $\text{CuO}_2$  sheet per a formula unit.<sup>1)</sup>

The role of doped holes in antiferromagnetic ordering of this system has not been understood yet, since the precise measurement of hole concentration,  $h=x+2\delta$ , is technically difficult. In this paper, we report the experimental result of the measurement of h as functions of the Sr concentration, x, and annealing conditions leading to the different excess oxygen,  $\delta$ . Here, the value of h has been determined by the coulometric titration method. The relation between h and antiferromagnetic ordering temperature,  $T_N$ , which was determined by temperature dependence of dc magnetic susceptibility is discussed from a view point of size scaling law.

### §2. Experimental

The samples of  $La_{2-x}Sr_xCuO_{4+\delta}$  for  $0 \le x \le$ 0.02 were prepared from high purity La<sub>2</sub>O<sub>3</sub> (99.99%), SrCO<sub>3</sub> (99.99%), and CuO (99.999%) powders. Starting materials were ground and pressed into pellets and preheated in air at 900°C for 24 hours. After heating, the products were quenched to room temperature, reground and pelletized again. They were annealed again to homogenize at 1050°C in air for 1 or 2 days. This process was repeated several times in order to obtain homogeneous samples. The powder sample was separated into two parts, one of which was annealed at 900°C in flowing  $N_2$  ( $P_{N_2}=1$  atm) with high purity of above 99.995% for 24 hours (N<sub>2</sub> treatment) and the other was done at 400°C in flowing  $O_2$  ( $P_{O_2}=1$  atm) with the same purity for 24 hours (O2 treatment) and both were cooled to room temperature at the rate of  $1 \sim 2^{\circ}$ C/min. The X-ray powder diffraction (XRD) measurements were performed to check the samples and all samples prepared were confirmed to be single phases having the K<sub>2</sub>NiF<sub>4</sub>-type structures with orthorhombic symmetry.<sup>4)</sup>

The value of h was determined by use of the coulometric titration technique using CuCl as

a reducing reagent.<sup>3,5)</sup> After the sample of  $La_{2-x}Sr_xCuO_{4+\delta}$  was dissolved into the solution of 1 N-HCl containing excess CuCl, holes in the sample existing as a form of [CuO]<sup>+</sup> easily oxidized Cu<sup>+</sup> in CuCl to Cu<sup>2+</sup>. The hole content was determined from the electric oxidation of the remaining Cu<sup>+</sup> by flowing constant current of 5 mA. We took sufficient care to remove  $O_2$  from the solution before titration because it also oxidizes Cu<sup>+</sup>.

The temperature dependence of the dc magnetic susceptibility  $\chi(T)$  was measured at 0.6 Tesla using a torsion magnetic balance.  $T_{\rm N}$  was assumed to be the peak temperature of  $\chi(T)$  curve, at which the three-dimensional long range antiferromagnetic ordering occurs resulting from a weak interaction between adjacent  ${\rm CuO_2}$  planes.<sup>1)</sup>

#### §3. Result and Discussion

The Sr concentration dependence of h determined from the coulometric titration analysis is shown in the upper part in Fig. 1. In the lower part of this figure,  $\delta$  vs. x curve is also presented. It is to be noted that h can be directly determined by the coulometric titration and that  $\delta$  can be determined from the equation,  $\delta = (h-x)/2$  by taking the nominal Sr content as x. For x=0, the difference of the oxygen contents between the samples with the  $O_2$  treatment and with the  $N_2$  treatment is about 0.003, which is consistent with the result of our thermogravimetric analyses.

The value of  $\delta$  obtained for each annealing condition gradually decreases with x. On the other hand, h increases with x. It is clear that the excess oxygen considerably contributes to the total hole concentration even in the samples with the  $N_2$  treatment.

The x dependence of  $T_N$  determined from  $\chi(T)$  is shown in the inset of Fig. 2 for each annealing condition. For the samples with the  $N_2$  treatment,  $T_N$  decreases drastically with x and the long range antiferromagnetic ordering disappears for x>0.02. In the same way, for the samples with the  $O_2$  treatment,  $T_N$  decreases drastically with x and goes down to 0 K around  $x\sim0.015$ . If we compare two samples having the same x value,  $T_N$  is always lower in the case of the  $O_2$  treatment than in the case of the  $N_2$  treatment, indicating  $T_N$  tends to

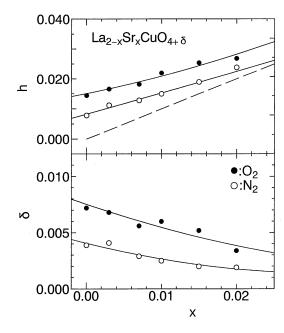


Fig. 1. (upper) Sr concentration, x, dependence of the hole concentration, h, determined from coulometric titration. (lower) Sr concentration, x, dependence of excess oxygen,  $\delta$ . Closed circles: the  $O_2$  treatment. Open circles: the  $N_2$  treatment. The solid lines are guides for the eyes. The broken line indicates the relationship for the system with no oxygen vacancy or no excess oxygen, i.e. h=x.

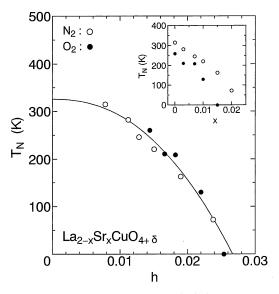


Fig. 2. The dependence of  $T_{\rm N}$  on the hole concentration, h. The solid line is the fit by the function in the text. (inset) The dependence of  $T_{\rm N}$  on the Sr concentration, x. Closed circles: the  $O_2$  treatment. Open circles: the  $N_2$  treatment.

decrease with introduction of the excess oxygen. Our  $T_N$  data are for the most part in agreement with the data of Cho et al.60 and Saylor and Hohenemser.7) However, we did not observe superconductivity down to 4 K in any samples of  $x \le 0.02$  with the  $O_2$  treatment, which differs from the results by Cho et al. We suppose that this difference is caused from the differences in preparing conditions. For instance, if the magnetic measurement is done for a sintered specimen, rather than for a powdered one, superconducting volume fraction is sometimes overestimated due to the weak coupling. Superconductivity observed by Cho et al. might be ascribed to a trace of superconducting phase formed due to inhomogeneity in the sample, though we do not know whether they measured a sintered specimen or not.

Next,  $T_N$  is plotted as a function of h in Fig. 2. One can notice that the decrease of  $T_N$  correlates significantly with h. The sharp decrease of  $T_N$  by doping small amount of holes can not be explained by the simple dilution model. In the non superconducting system of La<sub>2</sub>Cu<sub>1-y</sub>M<sub>y</sub>O<sub>4</sub> (M=Zn, Mg) and the n-type superconducting system of Pr<sub>2-y</sub>Ce<sub>y</sub>CuO<sub>4</sub>,  $T_N$  decreases more slowly with y.<sup>2,8)</sup>

Cho et al. pointed out that the decrease of  $T_{\rm N}$  with x (note that x is not equal to hole concentration in a strict sense as described above) showed power-law dependence. They suggested a microscopic phase separation where the doped holes are concentrated in hole-rich walls separating undoped domains. The microscopic phase separation has been found in much highly doped region from our measurement of nuclear quadrupole resonance.

The rapid collapse of the antiferromagnetic ordering can be explained from the finite size scaling theory.  $^{10)}$   $T_{\rm N}(h)$  curve in Fig. 2 can be fitted by the least-square analysis in the following expression

$$\frac{T_{\rm N}(h)}{T_{\rm N}(0)}=1-\left\{\frac{h}{h_{\rm c}}\right\}^n$$

where  $T_{\rm N}(0)$ =326 K,  $h_{\rm c}$ =0.0268 and n=2.27. This power-law dependence may be also explained by the finite size scaling theory. We suppose that there occurs the *microscopic* phase separation into antiferromagnetic and

non-antiferromagnetic regions due to the hole distribution. In this case, the finite domain means the one with the three-dimensional antiferromagnetic correlation (3dAFC). The reason why the domain size is finite comes from the fact that the antiferromagnetic correlation between the adjacent CuO<sub>2</sub> planes is considerably weaker than the one in the CuO<sub>2</sub> plane.<sup>11)</sup> Hole doping would reduce the size of the 3dAFC domain because, above  $T_N$ , the spinspin antiferromagnetic correlation length,  $\xi$ , rapidly decreases with increasing both x and  $\delta$ . 2,11,12) As far as the decrease of  $T_N$  is concerned, x and  $2\delta$  can be treated equivalently. The fitting in Fig. 2 leads the maximum value of  $T_N$  to be about 326 K with  $h\rightarrow 0$ . We conclude, therefore, that holes play a key role in disordering the quasi two-dimensional antiferromagnetic state in CuO<sub>2</sub> plane. Further study is desired in order to explain how doped holes destroy the antiferromagnetic ordering.

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  218
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